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Titanium Surface Modification by Oxidation for Biomedical Application

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ABSTRACT

Surface modification is a process that is applied to the surfaces of titanium substrates in order to improve the biocompatibility after implanting in the body. Two methods were used in the present work: Anodisation and gel oxidation. Anodisation was performed at room temperature in strong mineral acids (sulphuric acid (H₂SO₄) and phosphoric acid (H₃PO₄)), an oxidising agent (hydrogen peroxide (H₂O₂)), mixed solutions of the preceding three, and a weak organic acid mixture (β -glycerophosphate + calcium acetate). The parameters used in anodisation were: Concentrations of the electrolytes, applied voltage, current density, and anodisation time. Gel oxidation was carried out by soaking titanium substrates in sodium hydroxide (NaOH) aqueous solutions at different concentrations (0.5 M, 1.0 M, 5.0 M, and 10.0 M) at 60°C for 24 h, followed by oxidation at 400°, 600°, and 800°C for 1 h.

Conceptual models representing changes in the microstructure as a function of the experimental parameters were developed using the anodisation data. The relevant parameters were: Applied voltage, current density, acid concentration, and anodisation time:

- The model for anodisation using the strong acid (H_2SO_4) illustrates the growth rate of the film, identification of the threshold for the establishment of a consistent microstructure, and prediction of the properties of the film.
- For the oxidising agent (H₂O₂), two models were developed: Current-control and voltage-control, the applicability of which depends on the scale of the current density (high or low, respectively). These models are interpreted in terms of the coherency/incoherency of the corrosion gel, arcing, and porosity.
- The model for the strongest acid (H_3PO_4) is similar to that of H_2O_2 in current-control mode, although this system showed the greatest intensity of arcing and consequent pore size.
- Anodisation in mixed solutions uses Ohm's law to explain four stages of film growth in current-control mode. These stages describe the thickness of the gel, its recrystallisation, and the achievement of a consistent microstructure.

Anodisation in weaker organic acids allows the most detailed examination of the anodisation process. Both current density and voltage as a function time reveal the nature of the process in six stages: (1) instrumental response, (2 and 3) gel thickening, (4) transformation of the amorphous gel to amorphous titania, (5) recrystallisation of the amorphous titania, and (6) subsurface pore generation upon establishment of a consistent microstructure.

Gel oxidation was done at low and high NaOH concentrations followed by oxidation. Three models were developed to represent the gel oxidation process: (1) Low concentration, (0.5 M and 1.0 M NaOH), (2) Medium concentration (5.0 M NaOH), and (3) high concentration (10.0 M NaOH). For the low concentrations with increasing temperature, the model involves: (1) amorphous sodium titanate forms over a layer of amorphous anatase and (2) a dense layer of rutile forms. For the high concentrations with increasing temperature, the model involves: (1) amorphous sodium titanate forms over a layer of amorphous anatase and (2) a dense layer of rutile forms. For the high concentrations with increasing temperature, the model involves: (1) amorphous sodium titanate forms over a layer of amorphous anatase, (2) a dense layer of anatase forms and raises up the existing porous anatase layer, and (3) the dense and porous anatase layers transform to dense and porous rutile layers, respectively. The main difference between the two is the retention of crystalline sodium titanate in the higher NaOH concentration.

Anodised and gel oxidised samples subsequently were soaked in simulated body fluid in order to study the precipitation of hydroxyapatite in the absence and presence of long UV irradiation, which has not been investigated before. With the anodised surfaces, the porous and rough titania coating facilitated both the precipitation of hydroxyapatite and the attachment of bone-like cells. UV irradiation showed greatly enhanced hydroxyapatite precipitation, which is attributed to its photocatalytic properties. With the gel oxidised surfaces, the greatest amount of hydroxyapatite precipitation occurred with the presence of both anatase and amorphous sodium titanate. Rutile suppressed precipitation.

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CHAPTER 1

INTRODUCTION

Titanium and its alloys have been used widely in biomedical implants and dental applications for their excellent biocompatibility, superior mechanical properties, and corrosion resistance. These materials are used for the repair and reconstruction of damaged parts of the body, including replacements for hips, knees, teeth, and fixators for fractured bones.

In biomaterial applications, some general terms (Table 1.1) such as *toxic*, *bioinert*, *bioactive*, and *bioresorbable* are used to classify the material's tissue response [1]. Titanium is considered to be *non-bioactive* for implants in bones.

Material	Explanation	Explanation Tissue Response	
Torio	Material is toxic	The surrounding	Copper
TOXIC		tissue dies	
	Material is non-toxic and	A fibrous tissue of	Alumina, zirconia,
Bioinert	biologically inactive	variable thickness	titania, titanium
		forms	and it alloys
	Material is non-toxic and	An interfacial bond	Hydroxyapatite,
Bioactive	biologically active	forms, between the	bioactive glasses
		tissue and material	
	Material is non-toxic and	The surrounding	Tri-calcium
Bioresorbable	dissolves in surrounding	tissue replaces it.	phosphate
	enviroment		

 Table 1.1:
 Type of implant - tissue response [1].

For further improvement in biocompatibility, oxide surface modifications of titanium have been investigated. This is particularly relevant as all metallic titanium implants have a thin passivating oxide layer, so, in the past, these would have been implanted without the knowledge that the cellular response to the implant included the oxide component. Intentionally applied layers of titanium dioxide (TiO₂) produced by surface modification of titanium have emerged as important adjuncts to biomaterials owing to of their excellent chemical and physical properties.

 TiO_2 is known to have three natural polymorphs: rutile, anatase, and brookite [2]. Rutile, as the stable form of TiO_2 at ambient conditions, possesses unique semiconducting characteristics [3]. With regard to the chemical properties, anatase is an important phase since it is more reactive and thus more effective in forming apatite in simulated body fluid (SBF) [4]. It also is a more suitable form for photocatalytic activity than rutile [2]. Brookite normally is difficult to obtain during the ceramic processing [5].

Two methods, anodisation and gel oxidation, were used in the present work to form a titania layer on Ti substrates. Anodisation and gel oxidation are simple techniques that are useful at low temperature for producing TiO_2 layers on titanium substrates [6].

The TiO₂ layer on titanium was tested *in vitro* using simulated body fluid (SBF) using the method of Kokubo [4] and osteoblast bone-like cells. Previously, several researchers have conducted tests *in vitro* under normal conditions. However, no work has been carried out in the dark and under long-wave ultraviolet (UV) light, although TiO₂ is well known to react under ambient light. UV radiation can affect the TiO₂ implant surface, which is located near the skin (*e.g.*, dental implants and external bone-fracture fixator) and may change the properties of TiO₂ and its functions in the body in the long term.

CHAPTER 2

LITERATURE SURVEY

2.1 Titanium Dioxide (TiO₂)

Titanium dioxide (TiO₂) has been widely used in many industrial applications during the last few decades. Most of the applications are based on the special surfaces properties and catalytic properties. Due to these properties, TiO_2 has become the subject of many investigations for applications in optical, electrical and micro electronic, photonic, chemical and biomedical fields.

2.1.1 Uses

2.1.1.1 General Applications

 TiO_2 is commonly used as a thin film and powder form. The general uses of TiO_2 are summarised in Table 2.1.

2.1.1.2 Photocatalysis

Over the past several years, several applications of photocatalytic technology have been examined. Some in use applications of photocatalysis are shown in Figure 2.1[7].

Application	Explanation / Applications	References
White pigments	Used in paint, plastic, inks, paper, leather, textiles and cosmetic products	[7-11]
Metal oxide semiconductor field effect transistor	High dielectric constants ($\epsilon = 100$)	[9,11,12]
(MOSFET)		
Varistors	Used in ceramic and electric devices	[8]
Gas sensors	As humidity control To control the air/fuel mixture in car engines Utilized in determination of oxygen and CO at high temperatures (>600° C)	[8,9,11,13]
Biomaterials	Bone implant coatings As bone substituent As reinforcing mechanical support	[8,11]
Photo-assisted degradation of organic molecules	Purification of waste water Used in operating rooms in hospital Self-cleaning coating on car windshields	[7,8]
Anti cancer treatment	Photochemical treatment	[7]
As thin film optical interference coating	Antireflective coating Dielectric mirrors for lasers Metal mirrors with enhanced reflection Filters	[7,8,9,11]
As a protective coating	Corrosion resistant barriers	[8,9,13]
Photocatalysts (solar cells)	Used in the production of hydrogen and electric energy As anti-reflection coatings	[7,8,11,13]
Sunscreen	As UV absorber in sunscreen cream with high sun protection factors	[7,9,11]
Food	Foodstuff, food colouring (E-171)	[9,11]
Pharmaceuticals	As tablet coating, toothpastes	[9,11]
As catalysts	Selective reduction of NO_x to N_2 , Hydrogen production by gas shift production, CO oxidation by O_2 , H_2S oxidation to S, Reduction of SO ₂ to S by CO, NO ₂ storage	[9,11]
Used in fluxes and ceramics	Raw materials	[10]
Li-based batteries	Anatase form is used as anode material	[8,11]
Ultra-thin capacitors		[9]
Electrochromic devices	Thin film coating	[8,13]

Table 2.1: The applications of TiO_2 in various fields.



Figure 2.1: TiO_2 photocatalysis general applications [7].

Fujishima *et al.* [7] explained about the critical role of TiO_2 . By using TiO_2 in photocatalysis, the environment can be rendered clean and energy consumption can be decreased. Table 2.2 lists the various applications of TiO_2 as a photocatalysis [7].

Property/Function	Category	Applications		
	Materials for residential and office buildings	Exterior tiles, kitchen and bathroom components, interior furnishings, plastic surfaces, aluminium siding, building stone and curtains, paper window blinds		
Self-cleaning	Indoor and outdoor lamps and related systems	Translucent paper for indoor lamp covers, coatings on fluorescent lamps and highway tunnel lamp cover glass		
	Materials for roads	Tunnel wall, soundproofed wall, traffic signs and reflectors		
	Others	Tent material, cloth for hospital garments and uniforms and spray coatings for cars		
	Indoor air cleaners	Room air cleaner, photocatalyst-equipped air conditioners and interior air cleaner for factories		
Air cleaning	Outdoor air purifiers	Concrete for highways, roadways and footpaths, tunnel walls, soundproof walls and building walls		
Water purification	Drinking water	River water, ground water, lakes and water-storage tanks		
water purification	Others	Fish feeding tanks, drainage water and industrial wastewater		
Anti tumour activity	Cancer therapy	Endoscopic instruments		
Self-sterilising	Tiles to cover the floor and walls of operating rooms, silicone rubber for medical catheters and hospital garments and uniforms			

Table 2.2: The applications of photocatalysis [7].

2.1.1.3 **Biomedical Applications**

 TiO_2 is used in biomedical applications as fine particles, coatings and oxide films on the outer surface of biomaterials (Table 2.3). The surface of the Ti-based alloys forms TiO_2 layers in aqueous solutions in the human body and act as the interface for strong bonding with natural connective tissue [14]. The TiO_2 film also provides corrosion resistance and contributes to improve biocompatibility of the implant [8].

Fujishima *et al.* [15] implanted cancer cells under the skin of mice to cause tumours to form, and when the size of the tumours grew to about 0.5 cm, suspension containing fine particles of titanium dioxide was injected into it. After 3 days, the skin was cut open to expose the tumour and it was irradiated with ultraviolet (UV) and thus treatment clearly inhibited the tumour growth. However, this technique was not effective in stopping cancer which had grown beyond a certain size limit [15].

Form	Application / Explanations	References
	Oral implants	[8,16-18]
	Endosteal implants	
	Implant retained suprastructure	
Films on	Combined denture	
Metallic	Orthopaedic implants	[8,16,17]
Wietaine	Hip joint implants	
	Knee joint implants	
	Bone-fracture fixation (screws and plates)	
	External bone-fracture fixation	
Films on	Contact lenses	[7]
Polymers	Catheters	[7]
	Cancer treatment	[7]
	HeLa (cancer cells)	
	Pathogenic organism photodegradation	[11]
Powders	Escherichia coli	
	U 937 -Ten and 30 min of illumination in the presence	
	of TiO2 leads to 80% and complete killing of human U	
	937 monocytic leukaemia cells	

Table 2.3: Applications of TiO_2 in biomedical fields.

2.1.2 Crystal Structure and Phase Transformation

Titanium forms four well-defined oxides; monoxide (TiO), sesquioxide (Ti₂O₃), dioxide or titanic acid (TiO₂) called titania and pentoxide (Ti₃O₅) as shown in Figure 2.2. There are many differences in structural and chemical properties between its oxides

(Tale 2.4). From a practical standpoint, the dioxide (TiO_2) is the most important oxide [19].



Figure 2.2: Phase diagram of the Ti-O system taken from Samsonov [20]. The region Ti_2O_3 - TiO_2 contains Ti_2O_3 , Ti_3O_5 , seven discrete phases of the homologous series Ti_nO_{2n-1} (Magneli phases) and TiO_2 [8].

TiO₂ exists in nature in the form of minerals like anatase, brookite and rutile. Rutile is commonly found in nature; however, anatase and brookite are extremely rare. Generally, TiO₂ exist in an amorphous form at temperature below 350°C [9]. Above that temperature, anatase phase is formed and at temperatures greater than about 800°C, the most stable crystalline phase rutile, is formed [9,21].

According to Bokhimi [22], in most cases of TiO_2 synthesis, anatase is the main phase and brookite occurs as a minority phase, depending on synthesis conditions. The rutile phase is obtained by annealing anatase and brookite at temperature higher than 500°C [22]. The crystal structures of the three oxide forms can be discussed further in terms of orientation of the (TiO₂⁶) octahedral [11].

Duonoutry		TiO ₂	TO	T: O		
Property	brookite	anatase	rutile	110	$\Pi_2 O_3$	
Colour	dark brown	white	white	bronze	purple-violet	
Melting point, °C	-	-	1830-50	1737	2127	
Density (25°C), kg/m^3	4170	3900	4270	4888	4486	
Crystal structure	orthorhombic	tetragonal	tetragonal	Cubic	rhombohedr al [24]	

Table 2.4: Properties of titanium oxides at various oxidation states [12,23].

2.1.2.1 Anatase

Anatase which refers to the long vertical axis, was named by R.J. Hauy in 1801 from the Greek word 'anatasis' meaning 'extension' [11]. Anatase has a tetragonal crystalline structure (Figure 2.3a) [22] and is built up from octahedra that are connected at their edge (Figure 2.3b) [11].



Figure 2.3: Crystal structure of anatase [8,11,22].

2.1.2.2 Rutile

Rutile was discovered by Werner in Spain in 1803. Its name is derived from the Latin 'rutilus' meaning red. Rutile is the most stable form of TiO₂ [11]. Rutile has a crystal structure with tetrahedral symmetry (Figure 2.4a) [22]. Rutile is built up from octahedra that are connected predominantly at their edges (Figure 2.4b) [11].



(a)

Figure 2.4: Crystal structure of rutile [11,22].

2.1.2.3 Brookite

Brookite was discovered by A. Levy in 1825 at Snowen (Pays de Gales, England) and it was named in honour of the English mineralogist, H.J. Brooke [11]. Brookite has an orthorhombic crystalline structure. The crystal structure can be described as distorted octahedra with a titanium atom the centre and oxygen atoms in the vertices (Figure 2.5a) [22]. Brookite is built up from the octahedra that are connected at their corner and edges (Figure 2.5b) [11].



(a)

(b)

Figure 2.5: Crystal structure of brookite [11,22].

2.1.3 Thermodynamics and Phase Equilibrium

The thermodynamic phase stability in the Ti-O system is calculated based on calorimetric data [11]. Based on minor differences in the Gibbs free energy (4-20 kJ/mole) value between the three phases, the most stable phase is believed to be rutile at normal pressure and temperatures as compared to the other two phases. Particle size is also known affect to the phase stability due to its association with surface energy and surface stress. Thermodynamically, anatase is most stable at sizes less than 11 nm, brookite between 11 and 35 nm, and rutile at sizes greater than 35 nm. The transformation of anatase into rutile at room temperature is very slow and practically does not occur. At macroscopic scale, the transformations of bulk TiO₂ occur at temperatures more than 600° C.



The phase equilibria and crystal structures of the Ti-O system range between pure Ti and TiO₂ as shown in Figure 2.6. High solubility of oxygen in titanium (α Ti) at low temperature leads to formation of Ti₂O, Ti₃O and possibly Ti₆O. Ti₂O has the anti-CdI₂ structure with alternate oxygen layers vacant and additional vacancies randomly distributed in occupied layer [26-29]. Monoxides (γ TiO at high temperature)

structures are based on the NaCl structure. Four modifications of TiO have been identified, αTiO , βTiO , $\alpha Ti_{I-x}O$, and $\beta Ti_{I-x}O$ [25]. The special points of the Ti-O phase diagram are summarised in Table 2.5 and Table 2.6.

Phase	Com		Composition, at % O		Reaction type
$L + (\alpha Ti) \leftrightarrow (\beta Ti)$	5	13	8	1720 ± 25	Peritectic
$L \leftrightarrow (\alpha Ti)$		~24		1885 ± 25	Congruent
					Melting
$(\alpha Ti) + Ti_3O_2 \leftrightarrow Ti_2O$	33.3	40	33.9	~600	Peritectoid
$(\alpha Ti) + Ti_2O_3 \leftrightarrow Ti_3O$	~17	~25	~24.5	~500	Peritectoid
$L \leftrightarrow (\alpha Ti) + L$	~37	~31	~53	~1800	Monotectic (?)
$L + (\alpha Ti) \leftrightarrow \gamma TiO$	~55	31.4	34.5	1770	Peritectic
γTiO ↔ βTiO		-		~1250	Unknown
$\beta TiO \leftrightarrow \beta Ti_{1-x}O$		-			Unknown
$\beta \operatorname{Ti}_{1-x} O \leftrightarrow \alpha \operatorname{Ti}_{1-x} O$		-			Unknown
$(\alpha Ti) + \beta TiO \leftrightarrow \alpha TiO$	33.3	51	50	940	Peritectoid
$(\alpha Ti) + \alpha TiO \leftrightarrow Ti_3O_2$	32.4	50	40	920	Peritectoid
$\alpha \operatorname{Ti}_{1-x}O \leftrightarrow \alpha \operatorname{Ti}O + \beta \operatorname{Ti}_2O_3$	54.5	50	60	460	Eutectoid
$L \leftrightarrow \gamma TiO + \beta Ti_2O_3$	~57	54.5	59.8	1720	Eutectic
$L \leftrightarrow \beta Ti_2O_3$		60		1842	Congruent
$L + \beta Ti_2O_3 \leftrightarrow \beta Ti_3O_5$	63	60.2	62.5	1770	Peritectic
$\beta Ti_2O_3 \leftrightarrow \alpha Ti_2O_3$		60		~180	Unknown
$\beta Ti_3O_5 \leftrightarrow \alpha Ti_3O_5$		62.5		187	Unknown
$\gamma Ti_4O_7 \leftrightarrow \beta Ti_4O_7$		63.64		-123	Unknown
$\beta \operatorname{Ti}_4 \operatorname{O}_7 \leftrightarrow \alpha \operatorname{Ti}_4 \operatorname{O}_7$		63.64		-148	Unknown
$L \leftrightarrow \beta Ti_3O_5 + ?$	~64	62.5	-	~1670	Eutectic
$\beta Ti_3O_5 + \beta Ti_5O_9 \leftrightarrow \gamma Ti_4O_7$	62.5	64.29	63.64	~1500	Peritectoid
$L \leftrightarrow TiO_2$		66.7		1870	Congruent
$L \leftrightarrow (\beta Ti)$		0		1670	Melting point
$(\beta Ti) \leftrightarrow (\alpha Ti)$		0		882	Allotropic
					transformation

Table 2.5: The special points of the Ti-O phase diagram [25].

There are five polymorphs of TiO₂ that are anatase, brookite, TiO₂–II, TiO₂–III and rutile. Anatase and brookite are formed at low temperature and low pressure; TiO₂–II and TiO₂–III are formed from anatase or brookite under high pressure; and rutile is the stable phase at room temperature and pressure. The polymorphic transformations anatase \rightarrow rutile and brookite \rightarrow rutile do not occur reversibly [30,31]. The magneli phases are a series of discrete phases with stoichiometry Ti_nO_{2n-1} where $n \ge 2$. The magneli phases exist in between the monoxides (TiO) and dioxide (TiO₂). Roy and White have suggested that there exists discrete equilibrium phases for $n \le 99$ [32]. The

phases $\text{Ti}_n \text{O}_{2n-1}$ ($4 \le n \le 10$) have crystal structures derived from the rutile structure by crystallographic shear [25].

Phase	Composition, at % O	Pearson symbol	Space group	Structure-bericht designation	Prototype	
(βΤί)	0 to 8	cI2	Im3m	A2	W	
(aTi)	0 to 31.9	hP2	$P6_3/mmc$	A3	Mg	
Ti ₃ O	~20 to ~30	<i>hp</i> ~ 16	P-31c	-	-	
Ti ₂ O	~25 to 33.4	hP3	<i>P</i> ⁻ 3 <i>m</i> 1	-	Anti-CdI ₂	
γTi	34.9 to 55.5	cF8	Fm3m	<i>B</i> 1	CINa	
Ti ₃ O ₂	~40	$hP \sim 5$	P6/mmm	-	-	
βTiO	-	(a)	-	-	-	
αTiO	~50	<i>mC</i> 16	A2/ <i>m</i> or	-	-	
			$B^{*/*}$			
βTi _{1-x} O	~55.5	<i>oI</i> 12	<i>I</i> 222	-	-	
αTi _{1-x} O	~55.5	<i>tI</i> 18	I4/m	-	-	
βTi ₂ O ₃	59.8 to 60.2	hR30	<i>R</i> -3i	$D5_1$	αAl_2O_3	
αTi_2O_3	59.8 to 60.2	hR30	$R^{-3}c$	$D5_1$	αAl_2O_3	
βTi ₃ O ₅	62.5	(b)	-	-	Anosovite	
aTi ₃ O ₅	62.5	<i>mC</i> 32	C2/m	-	-	
α'Ti ₃ O ₅	(a)	<i>mC</i> 32	Cc	-	V_3O_5	
γTi_4O_7	63.6	aP44	<i>P</i> ⁻ 1	-	-	
βTi_4O_7	63.6	aP44	<i>P</i> ⁻ 1	-	-	
αTi_4O_7	63.6	aP44	P-1	-	-	
γTi ₅ O ₉	64.3	aP28	P-1	-	-	
βTi_6O_{11}	54.7	aC68	A ⁻ 1	-	-	
Ti ₇ O ₁₃	65.0	aP40	P-1	-	-	
Ti_8O_{15}	65.2	aC92	A ⁻ 1	-	-	
Ti_9O_{17}	65.4	aI52	<i>P</i> ⁻ 1	-	-	
Rutile	~66.7	tP6	$P4_2/mnm$	<i>C</i> 4	Rutile	
Metastable phase						
Anatase	-	<i>tI</i> 12	$I4_1/amd$	C5	Anatase	
Brookite	-	oP24	Pbca	<i>C</i> 21	Brookite	
High press	ure phase					
TiO ₂ -II	-	oP12	Pbcn	-	aPbO ₂	
TiO ₂ -III	-	~ <i>hP</i> 48	(c)	-	-	

Table 2.6: Ti-O crystal structure data [25].

(a) Cubic, (b) Monoclinic, (c) Hexagonal

2.1.4 **Properties of TiO**₂

The physical, optical, electrical and chemical properties of TiO_2 depend greatly on the amorphous or crystalline phase of the material [9]. The bulk properties of TiO_2 are shown in Table 2.7.

Bronosty	TiO ₂				
rioperty	brookite	anatase	rutile		
Space group	D_{2h}^{15} Pbca	$D_{4h}^{19} = I4_1 / amd$	D_{4h}^{14} $P4_2$ / mnm		
Lattice constant (nm)					
a	0.5346	0.3733	0.4584		
b	0.9166	-	-		
с	0.5135	0.9370	0.2953		
c/a	0.944	2.51	0.644		
Standard heat capacity, C_p^o ,	-	55.52	55.06		
298.15 J/(mol °C)					
Electron mobility, μ (cm ² /Vs)	-	~10	~1		
Band gap (eV)	-	3.2	3.0		
Refractive index					
n_g	2.809	2.5688	2.9467		
n_p	2.677	2.6584	2.6506		

Table 2.7: Bulk properties of the three main polymorphs TiO_2 (anatase, brookite and rutile) [8].

2.2.4.1 Physical Properties

The physical properties of TiO_2 depend on the amorphous or crystalline nature of the material. Amorphous TiO_2 has voids and relatively low density and does not have a crystallographic structure. The amorphous TiO_2 phase can be produced at low temperature $100^{\circ}C - 150^{\circ}C$ as a thin film deposit on substrates. However, TiO_2 thin films that are produced by chemical reaction form anatase at the lowest temperatures [9].

TiO₂ has three crystalline phases. In thin films, normally anatase and rutile are observed. Polycrystalline anatase film can be obtained from amorphous TiO₂ film by annealing at higher temperatures [9]. The transition from amorphous to crystalline anatase film occurs at ~ $300 - 365^{\circ}$ C and from anatase to rutile occurs at ~ $700 - 1100^{\circ}$ C. Both anatase and rutile have tetragonal crystallographic structure but rutile is more densely packed and thus denser than anatase. Amores *et al.* [33] have explained in detail about the phase transformation from anatase to rutile in the high temperature sintering process as shown in Figure 2.7. The proposed mechanism for the sintering and transformation of anatase into rutile comprises on:

- 1. The smallest particles coalesce, forming bigger particles.
- 2. The fractions of particles that are already large have been shown not to undergo sintering.
- 3. Heat evolved from the exothermic sintering process causes the local nucleation of rutile.
- 4. The conversion to rutile is also an exothermic process, leading to the transformation of the whole particle to rutile.



Figure 2.7: Proposed mechanism for the sintering and transformation of anatase into rutile. (adapted from [33]).

Anatase and rutile have different densities due to different packing arrangment. Table 2.8 shows the density of the various phases and thin films mentioned by Richards [9]. Density of TiO_2 thin film can be calculated by using the equation derived from the linear relationship between density and refractive index (Figure 2.8) [34].

Structure	Density (kg/m ³)	Notes	References
	2400	Porous film	[35]
Amomhous	3200 - 3650	Typical value	[36]
Amorphous	3600 - 3800	Films deposited with a high kinetic energy	[34]
Brookite	4170	Bulk	[8]
Anotoco	3900	Dulk	[8]
Anatase	3840	Duik	[9]
Rutile	4270	Bulk	[8]
	4260	Duik	[9]
	4090-4100	Thin film	[34,37]

*Table 2.8: Various density of TiO*₂ [9].

The equation of line in Figure 2.7 is

$$n_f = 0.42751\rho + 0.91933 \tag{2.1}$$

$$\rho = \frac{n_f - 0.91933}{0.42751} \tag{2.2}$$

where ρ and n_f are the TiO₂ film density (kg/m³) and refractive index respectively. The porosity of the thin film can be determined using equation 2.3 [38].

Porosity =
$$\frac{n_f^2 - 1}{n_b^2 - 1}$$
, (2.3)

where n_b is the refractive index of the bulk single crystal material. This value is an approximation due to the fact that firstly, both anatase and rutile crystals exhibit strong birefringence and secondly mixed anatase / rutile phases can exist [9].



Figure 2.8: Experimental data from previous researchers indicating that a linear correlation between TiO₂ film density and refractive index is observed over for a wide range of values. Ottermann and Bange [39], Fitzgibbons et al. [40], Bendavid et al. [34], Hass [41], and Ribarsky [42] were used. (Adapted from [34]).

2.2.4.2 Chemical Properties

 TiO_2 is relatively inert and has good corrosion resistance in a biological environment. TiO₂ film on titanium and titanium based alloys is widely used in biomedical and dental applications [43]. It exhibits stability and corrosion resistance that protects the metal from further oxidation in vitro. When in contact with body fluids that have pH values close to neutral, the materials show corrosion rates that are extremely low and difficult to measure experimentally.

In general, crystalline phases of TiO₂, anatase and rutile are much more chemically resistant than amorphous TiO₂ [9]. Richard also found that amorphous TiO₂ films are highly soluble in hydrofluoric acid (HF), but dense and polycrystalline films are insoluble. The chemical resistance of TiO₂ to sulphuric acid (H₂SO₄) is very dependent on the film preparation technique [44], but TiO₂ is insoluble in all other concentrate acids and bases [9]. Barksdale reported that TiO₂ is slightly soluble in H₂SO₄, HF and a few strong alkalis, but it is almost completely chemically inert after annealing at 1000°C [19]. Strong basic solutions such as sodium hydroxide (NaOH) and ammonium hydroxide (NH₄OH) are used in etching processes for TiO₂ films [9].

2.2.4.3 Optical Properties

The important optical properties of TiO_2 are refractive index and extinction coefficient. Refractive index indicates the extent to which a light beam is reflected when passing through as compared to the vacuum substance. The refractive index of a TiO_2 thin film is less than that of anatase or rutile crystal. Anatase and rutile are birefringent crystals and their refractive indices are calculated using mean refractive index n_{mean} for a randomly oriented polycrystalline thin film (Equation 2.4).

$$n_{mean} = \frac{2n_{\perp} + n_{\rm II}}{3} \tag{2.4}$$

where $2n\perp$ and $n_{//}$ are for oscillations perpendicular and parallel to the optical axis (axis of light can pass without undergoing double refraction), respectively [9]. The mean refractive indices for rutile and anatase (at $\lambda = 600$ nm) are 2.70 and 2.53 respectively [41,42].

Richard [9] presented information about the refractive index, optical bandgap and optical absorption in his work. The refractive indices for anatase single crystal from previous research and the rutile single crystal from Kim is shown in Figure 2.9. The

optical bandgap value of TiO₂ changes depending on its structure. The optical bandgap of amorphous TiO₂ is around 3.5 eV, while for anatase and rutile, it is about 3.2 eV and 3.05 eV, respectively. The optical absorption will increase with successive transformations from amorphous TiO₂ to anatase to rutile material. From Figure 2.10, anatase is seen to have an absorption edge with a lower steepness, which is attributed to the presence of excitons (bound state of an electron and a hole), and more imperfections and disorder than in anatase crystal. Figure 2.11 shows the exponential dependence of the absorption coefficient at 10 K for different polarisations when illuminated with UV light [9,45].



Figure 2.9: Published values for the refractive index of single crystal anatase, taken from Meyer and Pietsch [44]; Hass [41]; Fitzgibbons [40]; Kingery et al. [46]; Washburn [47] and Kim [48]. The dispersive curve for single crystal rutile from Kim is also given [48].



Figure 2.10: Fundamental absorption edge of anatase and rutile single crystal, measured at a temperature of 10 K (adapted from [45]).

The extinction coefficient k, play an attenuating role in the material. It is related to absorption coefficient α through this equation (Equation 2.5),

$$k = \frac{\alpha \lambda}{4\pi}, \qquad (2.5)$$

where k and α are extinction coefficient and absorption coefficient, respectively [9]. Richard [9] also mentioned that the extinction coefficient of a film can be increased by scattering of light by surface and volume in perfections, such as surface roughness, porous microstructure, and density fluctuations [9]. The term optical loss L is defined as

$$1 = R + T + L \tag{2.6}$$

with

$$L = A + S \tag{2.7}$$

where A is absorptance, S is the scattering component, R is reflectance and T is transmittance.



Figure 2.11: The exponential dependence of the absorption coefficient of single crystal anatase, measured at 10 K with light polarised in $E_{\perp}c$ and $E_{\parallel}c$ directions (adapted from [45]).

2.1.4.4 Photocatalytic Properties

The photocatalytic phenomenon has been studied extensively for several years by many researchers. Since TiO_2 is a semiconductor with a band gap of about 3.0 eV, UV light (with wavelengths shorter than ~400 nm) can excite pairs of electrons and holes [49]. The photogenerated electrons then react with molecular oxygen (O₂) to produce superoxide radical anions (•O₂-), and the photogenerated holes react with water to produce hydroxyl (•OH) radicals. These two types of reactive radicals then work together to decompose organic compounds. The longer the film is illuminated with UV light, the more organic material that is decomposed, such that even an oily stain on the surface would gradually disappear under UV light [7]. Overall, the photocatalysed reaction may be summarized as follows:

$$(Ox_1)_{ads} + (Red_2)_{ads} \xrightarrow{\text{semiconductor}} Red_2 + Ox_2$$

$$(2.8)$$

Depending on the sign of the change in Gibbs free energy (ΔG°) value of the reaction (2.8), the semiconductor-sensitised reaction may be an example of photocatalysis or photosynthesis, respectively [11]. For a semiconductor photocatalyst to be efficient,

the different interfacial electron processes involving e^- and h^+ must compete effectively with the major deactivation processes involving $e^- - h^+$ recombination, which may occur in the bulk or at the surface (Figure 2.12). Ideally, a semiconductor photocatalyst should be chemically and biologically inert, photocatalytically stable, easy to produce and to use, efficiently activated by sunlight, able to efficiently catalyse reactions, cheap, and without risks for the environment or humans. Titanium dioxide (with sizes ranging from clusters to colloids to powders and large single crystals) is close to being an ideal photocatalyst, displaying almost all the above properties. The single exception is that it does not absorb visible light.



Figure 2.12: Major mechanism occurring on semiconductors: (a) electron-hole generation; (b) oxidation of donor (D); (c) reduction of acceptor (A); (d) and (e) electron-hole recombination at surface and in bulk, respectively [48].

Both crystal structures anatase and rutile, are commonly used as photocatalysts, with anatase showing greater photocatalytic activity for most reactions. It has been suggested that this increased photoreactivity is due to anatase having slightly higher Fermi level, lower capacity to adsorb oxygen and higher degree of hydroxylation (*i.e.*, number of hydroxy groups on the surface) [50]. Reactions in which both crystalline phases have the same photoreactivity or rutile having a higher one [51] have also been reported. Furthermore, there are also studies which claim that a mixture of anatase

(70–75%) and rutile (30–25%) is more active than pure anatase alone. The disagreement of the results may be due to the intervening effect of various coexisting factors, such as specific surface area, pore size distribution, crystal size, and preparation methods, or it may be related to the way the activity is expressed.

The behavior of Degussa P25 commercial TiO_2 photocatalyst, consisting of an amorphous state together with a mixture of anatase and rutile in an approximate proportion of 80/20, is observed to be more active than both the pure crystalline phases together [52]. The enhanced activity arises from the increased efficiency of the electron–hole separation due to the multiphase nature of the particles. Another commercial TiO_2 photocatalyst Sachtleber Hombikat UV 100, consisting only of anatase, has a high photoreactivity due to fast interfacial electron-transfer rate. Water splitting is a special case, because band bending is necessary in order to oxidize water and large rutile particles (with a small surface area) are efficient [53].

2.2 Colour Properties of Titanium and TiO₂

2.2.1 Colour System

Colour is associated with visible light waves (wavelength distributions). Visible light is a very small part of electromagnetic energy; wavelengths within violet (400 nm) and red (700 nm) of the electromagnetic spectrum (Figure 2.13) [54,55]. The selective absorption of different amount of wavelengths within violet and red described as colour of objects. Wavelengths not absorbed are reflected or transmitted by objects and thus visible to observers [54].

The colour measurement for the scientific purpose is based on numerical representations or quantification of three colour response mechanisms in the human eye [54]. There are several methods of measurement:

- 1) RBG-based colour space
- 2) HSV and HLS colour space
- 3) CIE colour space



Figure 2.13: Visible light (wavelength, 400-700nm) as part of electromagnetic energy [55].

2.2.2 RGB-Based Colour Space



Figure 2.14: RGB space colour cube (Red corner is hidden from view)[55].

The most commonly used colour space is the RGB colour system (Figure 2.14). The RGB space is a three-dimensional colour cube with primary colour defined as (red,

green and blue), secondaries (cyan, magenta and yellow) and white. One corner of the cube is the origin for the RGB coordinate axes [56].

2.2.3 HSV and HLS Colourn Space

A second colour space commonly used is the HSV and HLS colour space that are transformations of RGB. The name HSV stand for (hue-saturation-value is synonymous with HSB, hue-saturation-brightness), and HSL (hue-saturation-lightness) colour spaces. The two spaces can be described as being single and double cones respectively (Figure 2.15) [57].



Figure 2.15: HSV and HSL colour space [55].

The components in HLS space are analogous, but not completely identical, to the components in HSV space. The hue component in both colour spaces is an angular measurement (represent by a disc). A hue value is indicated as in Table 2.9.

Primary colour		Secondary colour		
0°	red	60°	yellow	
120°	green	180°	cyan	
240°	blue	300°	magenta	

Table 2.9: Hue value and colours

The saturation component in both colour spaces describes colour intensity. A saturation value of 0 (in the middle of a disc) means that the colour is "colourless" (gray); a saturation value at the maximum (at the outer edge of a disc) means that the colour is at maximum "colourfulness" for that hue angle and brightness.

The value component in HSV describes the brightness. In both colour spaces, a value of 0 represents the absence of light, or black. In HSV space, a maximum value means that the colour is at its brightest. In HLS space, a maximum value for lightness means that the colour is white, regardless of the current values of the hue and saturation components [55]. Table 2.10 shows the difference between HSV and HSL system.

Table 2.10:The HSV (saturation and value) and HSL (saturation and lightness)[58]

HSV			HSL		
	0	colourless		0	colourless
Saturation	max	Saturation	100	Pure primary	
		colouiruilless		100	colour
Value	0	black	Lightnage	0	black
value	max	bright	Lightness	100	white

According to Harold [54], scientific colour is measured based on numerical representations of the three colours that response to human eye. The human eye has receptors (called cone cells) for short (*S*), middle (*M*), and long (*L*) wavelengths. In 1931, The Commission Internationale de L'Éclairage (CIE) has defined a system to compute a triple of numerical components to be the mathematical coordinates of colour space and was called 1931 CIE XYZ colour space [59,54]. In the CIE XYZ colour space, the tristimulus values are called *X*, *Y*, and *Z*, which are roughly red, green and blue, respectively.

It is convenient to work in a 2D colour space. This is commonly done by projecting the 3D colour space onto the plane X+Y+Z=1, yielding a CIE chromaticity diagram (Figure 2.16).

The projection is defined as:

$$x = \frac{X}{X + Y + Z}$$
$$y = \frac{Y}{X + Y + Z}$$
$$z = \frac{Z}{X + Y + Z} = 1 \quad x \quad y$$



Figure 2.16: CIE chromaticity diagram, (a) schematic and (b) Colour [55].

2.2.4 Colour Measurement

Colour measurement by visual is subjective and difficult to correlate the observation of different observer. Scientific colour measurement is based on numerical representation due to difficulty to identify constantly, even for the same observer. Quantitative technique to measure colour are designed to be used as standard measurement.

The opponent-colours (L,a,b-type) colour scale are used in this study to measure the colour of the samples. The L,a,b-type scale original developed and refined by Richard S. Hunter between 1942 and1958 [54]. In 1976, the CIE adopted another L,a,b-type

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