DEVELOPMENT OF POWDER METALLURGY ROUTE FOR PRODUCTION OF NOVEL FE-AL INTERMETALLICS FOR HIGH TEMPERATURE APPLICATIONS

FAZIMAH BT MAT NOOR

This thesis is submitted as a fulfillment of the requirements for the degree of Master in Mechanical Engineering

> Faculty of Mechanical and Manufacturing Engineering Universiti Tun Hussein Onn Malaysia

> > MAC 2009

To my mom; i want to thank you for all you have given to me and all you have done for me. Your love and enthusiasm for my pursuits gave me energy and encouragement when i needed it most.

To my lovely husband, your understanding, encouragement and love throughout this entire adventure have picked me up when i was down, and made the many great times even more wonderful.

ACKNOWLEGMENT

First and foremost, i am indebted to my main supervisor, Professor Dr. Ing. Ir. Darwin Sebayang and my co-supervisor, Dr. Pudji Untoro for lending me their knowledge and assisting me in completion of this work. Their guidance and direction helped me through many difficult times. I would also like to express my deep gratitude and appreciation to my readers, Dr. Abdul Kadir bin Masrom and Dr. Syahril DIC. Their comments and critique of this manuscript are deeply appreciated. I would also like to express my thanks to the faculty of Mechanical and Manufacturing Engineering, University Tun Hussein Onn Malaysia.



I would like at this time to acknowledge that my research was supported at University of Tun Hussein Onn Malaysia by Research and Innovation Centre with vote numbers of 0157 and 0265.

I wish to express my thanks to the materials science laboratory technician, Mr. Tarmizi, the metallurgy laboratory technician, Mr. Abu Bakar, the polymer and ceramic laboratory technician, Mr. Fazlanuddin and the material science laboratory technician at University Technology Malaysia, Mr Zainal for their assistance in setting the apparatus and equipment for samples preparation and samples analysis.

ABSTRACT

FeAl based intermetallic alloys are being proposed as engineering materials for high temperature applications due to their low density, low materials cost, low content of strategic elements and oxidation resistance. These intermetallic alloys are suitable for applications in aggressive and corrosive environments up to 900°C. However they may fail through loss of strength or gradually deteriorate through reaction with the surrounding atmosphere when exposed to temperature higher than 900°C. Therefore, the formation of a stable protective oxide scale or alumina on the surface is required to protect the underlying materials when exposed to high temperature. In this research, the FeAl alloys were produced by using powder metallurgy route which consisted of mechanical alloying process, cold compaction, sintering, hot compaction and surface treatment via ion implantation. The addition of reactive elements or their oxides such as Y, Y₂O₃ and CeO₂ by mechanical alloying or ion implantation method may improve their oxidation resistance through the enhancement of the alumina scale adhesion to the underlying alloys. Characterizations by using SEM and XRD were carried out before and after each process to investigate the microstructure, phase change and formation of the oxide layer. Cyclic oxidation tests were performed at 900°C and 1100°C to study the oxidation behavior of these intermetallic alloys. The results showed that the FeAl intermetallic alloys were successfully produced by mechanical alloying, hot compaction and surface treatment via ion implantation. The FeAl intermetallic with 3×10^{15} ion/cm² doses of yttrium implanted exhibited the lowest oxidation kinetics at 900°C while FeAl intermetallic with 1 wt% yttria and 9×10¹⁵ ion/cm² doses of yttrium implanted exhibited the lowest oxidation kinetics at 1100°C.



ABSTRAK



Aloi intermetalik berasaskan FeAl telah dicadangkan sebagai bahan kejuruteraan untuk aplikasi suhu tinggi kerana mempunyai ketumpatan dan kos bahan yang rendah, kandungan elemen strategik yang rendah dan ketahanan pengoksidaan yang baik. Walaubagaimana pun bahan ini berkemungkinan akan gagal melalui kehilangan kekuatan atau menjadi semakin lemah melalui tindak balas dengan persekitaran yang terdedah kepada suhu yang tinggi daripada 900°C. Oleh itu, pembentukan satu lapisan oksida atau alumina sebagai pelindung adalah sangat diperlukan untuk memelihara bahan dasar apabila terdedah kepada suhu tinggi. Dalam projek ini, intermetalik FeAl telah dihasilkan dengan menggunakan kaedah metalurgi serbuk yang terdiri daripada proses pengaloian mekanikal, proses pemadatan sejuk, pensinteran, pemadatan panas dan rawatan permukaan melalui implantasi ion. Penambahan unsur-unsur reaktif atau oksidanya seperti Y, Y2O3 dan CeO₂ secara pengaloian mekanikal atau implantasi ion boleh menguatkan lagi lekatan antara lapisan oksida dengan logam dasar dan seterusnya meningkatkan ketahanan pengoksidaan. Pencirian dengan menggunakan SEM dan XRD telah dijalankan sebelum dan selepas setiap proses untuk mengkaji perubahan mikrostruktur, perubahan fasa dan pembentukan lapisan oksida. Ujian pengoksidaan berkitar telah dilakukan pada 900°C dan 1100°C untuk mengkaji kelakuan pengoksidaan bahan ini. Hasil kajian menunjukkan bahawa intermetalik FeAl telah berjaya dihasilkan secara pengaloian mekanikal, pemadatan panas dan rawatan permukaan melalui implantasi ion. Intermetalik FeAl yang diimplan dengan 3×10¹⁵ ion/cm² menunjukkan kadar pengoksidaan terendah pada suhu 900°C manakala intermetalik FeAl dengan 1 wt% yttria dan diimplan dengan 9×10^{15} ion / cm² mempamerkan kadar pengoksidaan terendah pada suhu 1100°C.

CONTENTS

CHAPTER CONTENTS PAGE

TITLE	i
DECLARATION	ii
DEDICATION	iii
ACKNOWLEDGEMENT	iv
ABSTRACT	v
ABSTRAK	vi
CONTENTS	vii
LIST OF TABLES	х
LIST OF FIGURES	xi
LIST OF SYMBOLS	xv
LIST OF APPENDIXES	xvii

I

INTRODUCTION

1.1	Background of Study	1
1.2	Rational of Using Powder Metallurgy Route	4
1.3	Problem Statement	5
1.4	Objectives of Study	6
1.5	Scopes of Study	6

vii

Π

2.1	Introduction to Powder Metallurgy	8
2.2	Intermetallic Materials	13
2.3	FeAl Based Intermetallic Alloys	15
2.4	Powder Metallurgy Processing of FeAl	
	based Intermetallic Alloys	18
2.5	High Temperature Oxidation and Corrosion	
	Resistance of FeAl based Intermetallic Alloys	20
2.6	Surface Treatments/Modification via Ion	
	Implantation	28

METHODOLOGY Ш

MET	HODOLOGY	
3.1	Raw Materials	36
3.2	Mixing of Metal Powders by Mechanical	
	Alloying Process.	36
3.3	Compaction of Metal Powders	38
	3.3.1 Cold Compaction	39
	3.3.2 Hot Compaction	40
3.4	Sintering	41
3.5	Ion implantation Process for Surface Treatment	42
	3.5.1 Samples Preparation	43
	3.5.2 Determination of Ion Doses	43
3.6	Cyclic Oxidation Test	44
3.7	Characterization Methods	45
3.8	Flowchart	48

RESULTS AND DISCUSSIONS

IV

4.1	Ball Milling	49
4.2	As consolidated powders	63

4.3	Crystallite Size	71
4.4	High-Temperature Oxidation Test	71

V CONCLUSIONS AND RECOMMENDATIONS

5.1	Conclusions	92
5.2	Recommendations	94

REFERENCES APPENDIXES

95 99

ix

LIST OF TABLES

TITLE

PAGE

2.1	Comparison of powder metallurgy and competitive	
	metalworking techniques.	11
2.2	Properties of some intermetallic compounds.	14
2.3	Oxide-metal volume ratios.	21
2.4	The Chemical Compositions of Tested Alloys (at%).	25
3.1	The mixed compositions of tested Alloys (wt%).	37
3.2	Parameters for the ion implantation process.	44
4.1	Crystallite size of the FeAl samples with and without Y_2O_3	
	and CeO_2 addition, before and after the cold or hot compaction	
	by using Scherrer Equation Method.	71

LIST OF FIGURES

FIGURE NO.

TITLE

PAGE

2.1	Raw material utilization (percent utilization).	9
2.2	Powder metallurgy route chart.	12
2.3	The specific strength and operation temperature of	
	contemporary high-temperature materials.	14
2.4	Phase diagram of Fe-Al.	15
2.5	Crystal lattice structures.	16
2.6	Schematics of microstructural evolution during milling.	19
2.7	Oxidation rate law.	23
2.8	Mass gain vs. time for isothermal oxidation of doped and	
	undoped Fe-37Al.	26
2.9	Variation of mass gain as a function of exposure time during	
	oxidation.	27
2.10	Scanning electron micrographs showing cross-sectional views	
	of scales developed on samples oxidized at 900°C/100 h.	28
2.11	Scanning electron micrographs showing cross-sectional views	
	of scales developed on samples oxidized at 1100°C/100 h.	28
2.12	Schematic of the ion implantation process.	29
2.13	Basic processes of ion implantation.	30
2.14	Areas of layer depth (thickness) of various surface modification	
	and coating processes.	31



2.15	Cyclic oxidation kinetics of the bare and cerium implanted		
	AZ31 at 773 K in air.	32	
2.16	Weight gain vs. time curves of blank and yttrium-implanted		
	304 stainless steel during oxidation at 1273 K in air.	33	
2.17	Mass gain versus time of non-implanted and implanted AZ31		
	samples at 773 K in air for 96 h.	33	
2.18	Oxidation weight gain curves of (1) as-received zircalloy-4,		
	and after (2) $5x10^{16}$, (3) $1x10^{17}$ and (4) $2x10^{17}$ ion/cm ²		
	titanium ion implantation. Zircalloy-4 was oxidized in air at		
	500 °C for 100 min.	34	
3.1	Planetary Ball mill.	38	
3.2	Cold compaction die used and green compacts produced.	39	
3.3	Structural changes accompanying the preparation of a		
	compacted product.	40	
3.4	Tube furnace used for sintering process.	41	
3.5	Structural changes accompanying the preparation of a		
	sintered product.	42	
3.6	Schematic of the hot compaction technique.	42	
3.7	The TRIM simulation program used in determining the doses		
	of ion for the implantation.	44	
4.1	The SEM images of the as received aluminium powder and		
	iron powder.	50	
4.2	SEM images of the powders milled for 46 hours.	51	
4.3	The SEM images of the as milled Fe-45 at% Al with 1 wt%		
	Y_2O_3 addition and Fe-45 at% Al with 1 wt% CeO ₂ addition.	52	
4.4	EDS results of the as received (a) aluminium powder, (b) iron		
	powder, and as milled (c) Fe-45at%Al powders, (d) Fe-60at%Al		
	powders and (e) Fe-80at%Al powders.	54	
4.5	DTA traces for the as mixed (a) Fe-45 at% Al powders, and as		
	milled (b) Fe-45 at% Al powders, (c) Fe-60 at% Al powders and		
	(d) Fe-80 at% Al powders, (e) Fe-45 at% Al powders with 1 wt%		
	$\rm Y_2O_3$ addition, and (f) Fe-45 at% Al powders with 1 wt% $\rm CeO_2$		
	addition.	58	

xii

4.6	The XRD results of the as received (a) aluminium and	
	(b) iron powders and as milled (c) Fe-45at% Al.	
	(d) Fe-60 at% Al and (e) Fe-80 at% Al.	62
4.7	The XRD results of the as received (a) Fe-45 at% Al	
	powders with 1 wt% Y_2O_3 addition, and (b) Fe-45 at% Al	
	powders with 1 wt% CeO ₂ addition.	62
4.8	The differences between constrained and relaxed regions can	
	lead to cracking if the compact green strength is low or	
	springback is large.	64
4.9	Cold compacted samples (a) Fe-45at% Al and (b) Fe-80at% Al.	64
4.10	Hot compacted samples.	65
4.11	SEM images for the as sintered (a) Fe-45at%Al powders.	
	(b) Fe-60at%Al powders and (c) Fe-80at%Al powders.	65
4.12	SEM images for the as consolidated (a) Fe-45 at% Al,	
	(b) Fe-45 at% Al powders with 1 wt% Y ₂ O ₃ , and	
	(c) Fe-45 at% Al powders with 1 wt% CeO ₂ .	66
4.13	XRD results for the as consolidated (a) Fe-45 at% Al powders,	
	(b) Fe-60 at% Al powders and (c) Fe-80 at% Al powders.	69
4.14	Surface morphology for the un-implanted FeAl samples.	73
4.15	Surface morphology for the implanted FeAl samples.	74
4.16	Surface morphology of the un-implanted FeAl+Y ₂ O ₃ .	75
4.17	Surface morphology of the implanted FeAl+Y2O3 samples.	76
4.18	Surface morphology of the un-implanted FeAl+CeO ₂ .	77
4.19	Surface morphology of the implanted FeAl+ CeO ₂ .	78
4.20	Serious scale spallation near the sample edge oxidized	
	at 1100°C for the un-implanted (a) FeAl, (b) FeAl+Y ₂ O ₃ ,	
	and (c) $FeAl+CeO_2$.	79
4.21	Scale spallation near the sample edge oxidized at 1100°C	
	for the implanted (a) FeAl, and (b) $FeAl+Y_2O_3$.	80
4.22	The microstructure at the scale spallation area at (a) 100x	
	magnification, (b) 500x magnification, and	
	(c) 1000x magnification.	81

xiii

4.23	The microstructure of the oxide scale form in the FeAl sample	
	after oxidized at (a) 900°C, and (b) 1100°C.	82
4.24	The SEM images at the cross-section area for (a) un-implanted	
	FeAl, and (b) implanted FeAl after oxidation at 900°C.	83
4.25	The SEM images at the cross-section area for (a) non implanted	
	FeAl, and (b) implanted FeAl after oxidation at 1100°C.	84
4.26	The XRD results for the samples oxidized at 900°C.	85
4.27	The XRD results for the samples oxidized at 1100°C.	86
4.28	Variation of mass gain as a function of exposure time during	
	Oxidation.	91

xiv

LIST OF SYMBOLS

xv

Fe	-	Ferum/Iron
Al	-	Aluminium
FeAl	-	Ferum-Aluminium
Y_2O_3	-	Yttria
CeO ₂	-	Ceria
Al_2O_3	-	Aluminium Oxide/ Alumina
Cr_2O_3	-	Chromia
SiO_2		Silicon Oxide
Y	-	Yttrium
Ce	-	Cerium
La	-	Lanthanum
XRD	-	X-Ray Diffraction
SEM	511	Scanning Electron Microscope
EDX	<u>r</u>	Energy Dispersion X-ray
DTA	-	Differential Thermal Analysis
PM	-	Powder Metallurgy
SHS	-	Self-propagation High temperature Synthesis
RE	-	Reactive Element
II	-	Ion Implantation
TRIM	-	Transport of Ion in Matter
h	-	Hour
Wt%	-	Weight Percentage
At%	-	Atomic Percentage

FCC-Face Cubic CentreBCC-Body Cubic Centre



xvi

LIST OF APPENDIXES

APPENDIXES.

TITLE

PAGE

А	Calculation of Crystallite Size by Scherrer Equation Method	100	
В	Presented Paper 1: Development of High-Temperature Materials		
	Fe-Al based Alloys by using Powder Metallurgy. The 9 th		
	International Conference on Quality in Research (QiR),		
	6-7 September 2006, Depok, Indonesia.	102	
С	Presented Paper 2: Oxidation Behavior of Yttrium Implanted		
	Fe-Al based Alloys at 1100°C. 3 rd Colloquium on Postgraduate		
	Research Colloquium on Materials, Minerals and Polymers		
	2007 (MAMIP 2007), 10-11 April 2007, Penang.	103	
D	Presented Paper 3: Powder Metallurgy Route for Production of		
	Novel FeAl Intermetallic for High Temperature Application.		
	World Engineering Congress 2007, 5-9 August 2007,		
	Penang, Malaysia.	104	
E	Presented Paper 4: Oxidation behavior of Fe-45Al Intermetallics:		
	The Effects of Y_2O_3 and CeO_2 on cyclic Oxidation Kinetics.		
	International and INCCOM-6 Conference, 12-14 Dec 2007,		
	Kanpur, India.	105	
F	Accepted Abstract: The Effect of Reactive Elements Addition		
	and Consolidation Methods on The Fe-45Al Crystallite Size,		
	International Conference and Exhibition on Structural Integrity		
	and Failure, SIF 2008, 9-11 July 2008, Australia.	106	

xvii

CHAPTER I

INTRODUCTION

1.1 Background of Study



The need for materials with high temperature capability in industries such as electric power generation, transportation and materials production/processing has increased dramatically since early 1900s. Process efficiency increases with operating temperature and early attempts to improve efficiency by raising temperatures were not always successful. Materials with the necessary capability were rarely available and the importance of high temperature materials in determining equipment performance and reliability was gradually appreciated. From the mid 1900s accelerating effort has been directed towards increasing the temperature capability of existing materials systems and developing new materials types. Understanding the material behavior and control of component manufacture to ensure the desired behavior have been key elements of these activities for all materials systems.

The requirement to operate at progressively higher temperatures will remain an ongoing need for the foreseeable future. Many industries will benefit from increased operating temperatures. As for example, in electricity generation, the efficiency of ultra-supercritical pulverized coal power plant can be increased from the current 47% to 50% if steam parameters can be increased from 290-bar/580°C to 325-bar/625°C. This will give major saving in fuel and consequent environmental benefits. Materials with higher temperature capability are essential if these, and many other objectives are to be met [1].

High temperature materials research in the metals and alloys area is still an extremely important field, and new alloy and composite systems are continually being developed for new applications. High temperature materials can be defined in several different ways, all of which are somewhat arbitrary. The least arbitrary definition is based on the maximum temperature used as a proportion of the melting temperature. In materials science and technology, high temperature can be defined as a temperature equal to, or greater than, approximately two-thirds of the melting point of a solid. High-temperature intermetallics have been vigorously studied since the early 1950s for various applications such as for the aerospace and powergeneration industries. In this study, high temperature materials are taken to be those materials that are used specifically for their heat-resisting capabilities, such as strength or resistance to oxidation above 900°C. Many efforts have been made to improve materials high temperature oxidation resistance. There are two methods in improving the oxidation resistance, one is alloying and the other is surface treatment.



Intermetallic is a class of materials which formed by the combination of two or more metal elements, generally (but not always) falling at or near a fixed stoichiometric ratio and ordered on at least two or more sublattices. Aluminides, like NiAl, TiAl and FeAl alloys are one of the most materials considered for high temperature applications. Serious research on high temperature intermetallics began in the early 1950's and increased significantly around 1970 because of their perceived potential in aerospace. With weight saving being a key requirement, early work concentrated on the aluminide intermetallics based on nickel and titanium. Subsequent intermetallics such as Fe₃Al have been developed because of their potential benefits in replacing steels in various high temperature applications. They

2

show excellent oxidation resistance, which is achieved by the formation of a protective Al_2O_3 -scale.

Iron aluminides or FeAl are of great interest because of their many unique properties. The most exciting of these properties include low density, lower cost because iron and aluminium are the most abundantly available elements and excellent resistant to high-temperature oxidation [2]. They contain enough aluminium to form thin films of aluminium oxides in the oxidizing environments that are often protective. They posses relatively high specific strengths and suitable mechanical properties at elevated temperatures. However, these alloys only suitable for applications in aggressive and corrosive environments up to 900°C. Besides, the commercial uses of these compounds have been seriously hindered by deficiencies in their mechanical properties, mainly the poor ductility at room temperature and the inadequate creep resistance at high temperature. Therefore, the FeAl intermetallics alloys have undergone extensive development in the recent years exclusively for high temperature applications [3]. Powder metallurgy is one of the method used to introduce a dispersion of Y₂O₃ (1% in weight) via mechanical alloying process which can increase the mechanical properties of the FeAl intermetallics alloys at room temperature and improve the alloys creep resistance and high temperature strength.



The oxidation resistance might become a limiting factor for component design in the application at high temperature. These had become the main problem for heat-resistant application [4]. Metallic materials are protected against high-temperature oxidation by the formation of protective oxide scales such as Cr_2O_3 , SiO_2 or Al_2O_3 , which possess sufficiently low growth rates to prevent rapid component degradation. In this study, FeAl based alloys had been developed by using the powder metallurgy methods. Powder metallurgy methods have an advantage with respect to microstructure control, materials used, product homogeneity and mass production. The exploring and using of powder processing techniques will lead to improve mechanical properties due to their smaller grain size [5]. Decreasing the grain size will increase the material ductility [6][7].

Mechanically alloyed materials also have excellent oxidation and hot corrosion resistance. The increased resistance to oxidation-sulfidation attack is due to the homogeneous distribution of the alloying elements and the improved scale adherence due to the dispersoid itself. While concerning improvements in alloy corrosion resistance at high temperature, the most interesting chemical elements to incorporate into iron-base alloys are oxygen-active elements, such as yttrium, or rare earths, such as cerium or hafnium [8].

Surface treatment via ion implantation technique can modify the corrosion behavior and surface mechanical properties such as hardness and fatigue resistance of the alloys. Besides, this ion implantation will also lowers the coefficient of sliding friction and modifies the surface chemistry [9]. The using of the ion implantation technique will allows the introducing of a controlled yttrium concentration into the sample. Cyclic oxidation test and characterization of the samples are carried out to study the effect of each processes used in the powder metallurgy methods to the materials microstructure, phase transformation, and material properties or performance by using Scanning Electron Microscope (SEM), Energy Dispersion X-Ray (EDX) and X-Ray Diffraction analysis (XRD).

1.2 Rational of Using Powder Metallurgy Route

Intermetallic alloys are invariably made of alloying constituents having large difference in melting temperature and density leading to segregation of components during formation. Such alloys are apt to have poor room temperature ductility. Efforts were made in various laboratories to prepare these materials by using conventional melting and powder processing approaches. Both these techniques employ pure components for melting. Conventional melting is usually carried out either by arc melting or by induction melting techniques under air, vacuum or controlled atmosphere. However, conventional melting encounters a host of problems related to melting. Firstly, due to large difference in melting temperature and density of iron and aluminium, homogeneous melting is not achieved. Secondly, adding aluminium to molten iron causes a large temperature rise of the molten bath. Such a sudden rise in temperature is dangerous for operators and results in melt oxidation, longer holding time prior to pouring and a potential for missing the target chemistry because of oxidation of alloying elements. Therefore, in this work the mechanical alloying process in the powder metallurgy route had been used to produce the FeAl intermetallics materials since this process capable of producing many high temperature melting intermetallics with homogenous structure and composition easily.

1.3 Problem Statement



Iron aluminides or Fe-Al based intermetallic alloys suitable for applications in aggressive and corrosive environments up to 900°C. However they may fail through loss of strength or gradually deteriorate through reaction with the surrounding atmosphere at temperature higher than 900°C. Therefore, the formation of a stable protective oxide scale on the surface is required to protect the underlying materials. The addition of 0.1 wt% to 2 wt% reactive elements or its oxide such as Y, Y₂O₃ and CeO₂ by mechanical alloying or treating the surface by implantation methods may improve their oxidation resistance through the enhancing of the alumina scale adhesion to the underlying metals. The oxide layer or scale formed tends to spall easily and the oxidation rate will increase if the amount is more than 2 wt% or less than 0.1 wt% [10].

1.4 **Objectives of Study**

The objectives of this research are:

- i. To develop FeAl based intermetallics alloys for high-temperature applications by using powder metallurgy method.
- ii. To study the effect of reactive elements addition such as yttria (Y2O3) and cerium dioxide (CeO₂) on the FeAl microstructure and oxidation behavior at 900°C and 1100°C.
- iii. To study the effect of yttrium ion implantation on the FeAl oxidation behavior at 900°C and 1100°C.
- KAAN TUNKU TUN AMINA iv. To determine of the optimum way of developing novel FeAl based alloy for high temperature application based on the results analyzed.

1.5 Scopes of Study

The scopes of this research are:

- i. Development of high temperature material FeAl based alloy with and without addition of reactive elements such as Y₂O₃ and CeO₂ by using mechanical alloying process and followed by powders consolidation.
- ii. Characterization of the as milled powders and as consolidated powders by using the Scanning Electron Microscope (SEM), Energy Dispersion X-ray (EDX), X-Ray Diffraction analysis (XRD) and Differential Thermal Analysis (DTA).

- iii. Implementation of surface treatment via ion implantation with different doses of yttrium ion, 3×10^{15} ion/cm², 6×10^{15} ion cm² and 9×10^{15} ion/cm².
- Implementation of cyclic oxidation at 900°C and 1100°C for 200 hours oxidation time.
- v. Microstructure and phase analysis before and after the implementation of ion implantation and cyclic oxidation by using SEM and XRD.
- vi. Determination of the optimum way of developing FeAl based alloy with and without reactive elements addition and yttrium ion implantation for high temperature application based on the results analyzed.

7

CHAPTER II

LITERATURE REVIEW

2.1 Introduction to Powder Metallurgy



It is well-known that materials with the same nominal composition can be produced in different ways, from classical melting and casting, practiced by conventional metallurgy, via powder consolidation by powder metallurgy methods, mechanical alloying, combustion synthesis with thermo-mechanical treatment, etc. Each technological route produces material having different microstructure, different concentrations and types of defects and therefore totally different properties [11].

Generally, powder metallurgy (PM) applications fall into two main groups. The first group consists of components difficult to make by any other production method, such as those components made from tungsten, molybdenum, or tungsten carbide. Porous bearings, filters, and many types of hard and soft magnetic components are also produced exclusively by the PM process. The second group of applications consists of components for which PM produces near-net shape components and is a cost-effective alternative to machined components, castings and forgings. Examples are automotive components such as clutch plates, connecting rods, cam shafts, and planetary gear carriers. PM components are used in a variety of

REFERENCES

- Meetham, G.W. and Van De Voorde, M.H. (2000). "Materials for High Temperature Engineering Applications." Berlin: Springer. 155-169.
- Sikka, V.K., Wilkening, D., Liebetrau, J., Mackey, B. (1998). "Melting and Casting of Fe-Al Based Cast Alloy." *Materials Science and Engineering*. A258. 229-235.
- [3] Babu, N., Balasubramaniam, R., and Ghosh, A. (2001). "High-temperature oxidation of Fe₃Al-based iron aluminides in oxygen." *Corrosion science*.
 43. 2239-2254.
- [4] Jiraskova, Y., Schneeweiss, O., Milickar, K., Svoboda, M., Prochazka, I.
 (2000). "Surface Microstructure Changes in the Mechanically Strained Fe-Al Based." *Materials Science and Engineering*. A293. 215-222.
- [5] Buscho, Cahn, Flemings, Ilschner, Kramer, and and Mahajan. (2000).
 "Intermetallics: Ion Aluminides." Encyclopedia of Materials Science and Technology. New York: Elsevier. 4170.
- [6] Deevi, S.C. (2000). "Powder Processing of FeAl Sheets by Roll Compaction." Intermetallics. 8. 679-680.
- Buscho, Cahn, Flemings, Ilschner, Kramer and Mahajan (2001).
 "Intermetallics: Ion Aluminides." Encyclopedia of Materials Science and Technology. New York: Elsevier. 4201.
- [8] Caudron, E., Buscail, H., Haanapel, V.A.C., Jacob, Y.P., Josse-Courty, C. and Stroosnijder, M.F. (2000). "Surface Modifification Effects Induced by Yttrium Implantation on Pure Iron and Varipus Steels." *Applied Surface Science.* 153. 156-165.
- Bloor, Brook, Flemings, Mahajan (1994). "Ion Implantation." The Encyclopedia of Advanced Material. Oxford: Pergamon. 1175.

- [10] Quaddakkers (2000). "Bath to Bath Variations in the Oxidation Behavior of Alumina Forming Fe-based Alloys." *Materials and Corrosion*. 51.350-357.
- [11] Godlewska, E., Szczepanik, S., Mania, R., Krawiarz, J., Kozin^{*} ski, S. (2003).
 "FeAl materials from Intermetallic Powders." *Intermetallics*. 11. 307-312.
- [12] US Department of Energy (2001). "Powder Metallurgy and Particulate Materials: Vision and Technology Roadmap." Office of Industrial Technology.
- [13] German, R.M. (1998). "Powder Metallurgy of Iron and Steel." New York: John Wiley. 128-180.
- [14] Yang, M.R., and Wu, S.K. (2003). "Oxidation Resistance Improvement of TiAl Intermetallics using Surface Modification." *Bulletin of the college of Engineering*, N.T.U. 3-19.
- [15] Askeland and Phule (2003). "The Science and engineering of Materials." 4th edition. USA: Thomson. 454.
- [16] Montealegre, M., Strehl, G., Gonzalez-Carrasco, J.L., and Borchardt, G.
 (2005). "Oxidation behaviour of novel ODS FeAlCr Intermetallic alloys." *Intermetallics*. 13, 896-906.
- [17] Buscho, Cahn, Flemings, Ilschner, Kramer, Mahajan. "Encyclopedia of Materials Science and Technology." New York: Elsevier. Vol 5.4170.
- Bora, A., Singha, P.P., Robi, P.S., and Srinivasan, A. (2004). "Powder metallurgy processing of ruthenium aluminum alloys." *Journal of Materials Processing Technology.* 153–154. 952–957.
- [19] Chan, C.D.N., Huvier, C., and Dinhut, J.F. (2001). "High Temperature Corrosion of some B2 Iron Aluminides." *Intermetallis.* 9. 817-826.
- [20] Morris, D.G., Munoz Morris, M. A., and Chao, J. (2004). "Development of High Strength, High Ductility and High Creep Resistant Iron Aluminide." *Intermetallics.* 12. 821–826.
- [21] William F. Smith (1996). "Principles of Materials Science and Engineering". New York: Mc Graw-Hill. 768.
- [22] Fontana, M. G. (1986). "Corrosion Engineering". Third Edition. International Student Edition. New York: McGraw-Hill. 505-544.

- [23] Zhou, Y., Peng, X., and Wang, F. (2006). "Cyclic oxidation of alumina Forming Ni–Al Nanocomposites With and Without CeO₂ Addition." *Scripta Materialia*. 55. 1039–1042.
- [24] Riffard, F., Buscail, H., Caudron, E., Cueff, R., Issartel, C., Perrier, S. (2006).
 "The influence of Implanted Yttrium on the Cyclic Oxidation Behaviour of 304 Stainless Steel." *Applied Surface Science*. 252. 3697–3706.
- [25] Ailor, W.H. (1971). "Handbook of Corrosion Testing and Evaluation." New York: John Wiley. 291-356.
- [26] Echsler, H., Alija Martinez, E., Singheiser, L., and Quadakkers, W.J.
 (2004). "Residual Stresses in Alumina Scales Grown on Different Types of Fe–Cr–Al Alloys: Effect of Specimen Geometry and Cooling Rate." *Materials Science and Engineering*. A 384.1–11.
- [27] Pedraza, F., Grosseau-Poussard, J.L., Dinhut, J.F. (2005). "Evolution of Oxide Scales on an ODS FeAl Intermetallic Alloy During High Temperature Exposure in Air." *Intermetallics.* 13. 27–33.
- [28] Grosdidier, T., Ji, G., and Bozzolo, N. (2006). "Hardness, Thermal Stability and Yttrium Distribution in Nanostructured Deposits Obtained by Thermal Spraying from Milled—Y2O3 Reinforced—or Atomized FeAl Powders." *Intermetallics.* 14. 715–721.
- [29] Haynes, J. A., Pint, B. A., Porter, W. D., and Wright, I. G. (2004).
 "Comparison of Thermal Expansion and Oxidation Behavior of Various high-temperature coating materials and superalloys." *Materials at High Temperatures*. 21. 87–94.
- [30] Xu, C.H., Gao, W. and Gong, H. (2000). "Oxidation Behaviour of FeAl Intermetallics: The Effects of Y and/or Isothermal Oxidation Kinetics." *Intermetallics*. 8. 769-779.
- [31] Montealegre, M.A, Gonzalez, J.L. Gonzalez-Carrasco (2003).
 "Influence of the Yttria Content of Oxidation Behaviour of the Intermetallic Fe40Al Alloy." *Intermetallics.* 1. 169-175.
- [32] Dang. Ngoc Chan, C., Huvier, H., and Dinhut, J.F. (2001). "High Temperature Corrosion of some B2 Iron Aluminides." *Intermetallics*. 9. 817-826.

- [33] Tomislav Filetin (2001). "An Overview of the Development and Application of Advanced Materials." *Croatian Welding Society, Zagreb.* 1-46.
- [34] Ding, W., Wang, X., Zeng, X., Wu, G., Yao, S., Lai, Y. (2007). "Cyclic Oxidation Behaviour of Cerium Implanted AZ31 Magnesium Alloys." *Materials Letters*. 61.1429–1432.
- [35] Wang, X. (2007). "The Effect of Y-Ion Implantation on the Oxidation of AZ31 Magnesium Alloy." *Materials Letters*. 61. 968–970.
- [36] Zhang, X.Y. (2006). "Effect of Titanium Ion Implantation on the Oxidation Behaviour of Zircalloy-4 at 500°C". Vacuum. 80. 1003–1006.
- [37] Raja Mohammad, R.S.A. (2004). "Study on the Possibility of Producing FeAl Intermetallic by using Mechanical Alloying". KUiTTHO: Tesis Sarjana Muda.
- [38] Skoglund, H., Wodel, M.K., and Karlsson, B. (2004). "Processing of Finegrained Mechanically Alloyed FeAl." *Intermetallic.* 12. 977-983.
- [39] Munoz-Morris, M.A., Garcia Oca, C., and Morris, D.G. (2003).
 "Microstructure and Room Temperature Strength of Fe-40Al Containing Nanocrystalline Oxide Particles." *Acta Materialia*. 51 . 5187-5197.
- [40] Zhu, S.M., and Iwasaki, K. (1999). "Characterization of Mechanically Alloyed Ternary Fe–Ti–Al Powders." *Materials Science and Engineering*. A270. 170–177.
- [41] Suryanarayana, C. (2004). "Mechanical Alloying and Milling." New York: Marcel Dekker.1-184.